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# Urea Assisted Combustion Synthesis of $\text{LiFePO}_4/\text{C}$ Nano composite Cathode Material for Lithium Ion Battery Storage System

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## Abstract

In the present work, we demonstrated economical way to synthesize nano crystalline  $\text{LiFePO}_4$  Olivine structured  $\text{LiFePO}_4/\text{C}$  nanocomposite using urea as fuel and sucrose as carbon source were calculated using the total oxidizing and reducing valencies of the components which provide as numerical coefficients for stoichiometric balance so that the equivalent ratio is unity followed by heat treatment at 600 °C for 6 - 10 h, under Ar (90%) and H<sub>2</sub> (10%) gas atmosphere. The resultant products are characterized by XRD to know the phase formation and for estimation of the crystallite size. The XRD patterns were collected in steps of 0.02° over the range of  $10^\circ \leq 2\theta \leq 80^\circ$  at room temperature with a constant counting time of 0.1 s per step, From XRD all the powders showing the nano sized crystallite sizes in the range of 40-45nm. The morphologies of the powders were investigated by a scanning electron microscopy (SEM) all samples indicate the agglomeration of powder with highly porous nature in all the samples. The synthesized olivine powders were subjected to thermogravimetric / differential scanning calorimetry (TG-DSC) as the reference material in the temperature range of 25–1000°C with a heating rate of 10°C/min in air and nitrogen atmosphere. Raman spectra were obtained with 514.5 nm Ar-Ion laser radiation at a laser power of 150 mW in the range of 100- 2000 $\text{cm}^{-1}$  for a total of 10 scans to increase the signal-to-noise ratio and with 4  $\text{cm}^{-1}$  spectral resolution were studied.

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*Keywords:*  $\text{LiFePO}_4$ , Urea, X-Ray Diffraction analysis (XRD), Scanning Electron Microscopy(SEM), thermogravimetric/ differential scanning calorimetry (TG-DSC), Raman spectrum.

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## 1. Introduction

With the entrance into the 21<sup>st</sup> century, our society has changed into an information-rich society. All kinds of information fill our daily life up. Lithium ion battery has become the battery of choice in large scale applications, because of their high energy density. The cathode material, plays a virtual role in lithium ion battery. Li-ion batteries are often selected as a power supply for these high-tech products. LiFePO<sub>4</sub> is one of the most promising candidates among the large number of Fe-related compounds due to its high energy density (theoretical capacity: about 170mAhg<sup>-1</sup>; working voltage: 3.5 V). Low electronic conductivity has been considered as the main problem of LiFePO<sub>4</sub> [1]. During charge process, the extraction of lithium from LiFePO<sub>4</sub> is accompanied by a direct transition to FePO<sub>4</sub>, [2] in which Fe<sup>2+</sup> ions are oxidized to Fe<sup>3+</sup>. Both LiFePO<sub>4</sub> and FePO<sub>4</sub> are poor electronic conductors [3] since there is no mixed-valent cations in these two phases, thereby causing poor electronic conductivity of LiFePO<sub>4</sub> electrode [4]. Thus, Li ions can only be partially extracted / inserted at room temperature at modest rates [5]. In order to obtain high power output [6], many methods have been carried out to overcome this limitation [7], including carbon coating [8] or Ag coating [9] [10] etc.

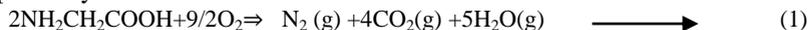
In this paper we are presenting an economical and simple combustion bottom-up method to synthesize LiFePO<sub>4</sub>/C Nano composite material as cathode for lithium ion battery. In the present study, we have developed a urea-assisted combustion method, with sucrose used in the initial stages as carbon source as well as a reducing agent. This process is a simple and inexpensive route for obtaining LiFePO<sub>4</sub>/C material and has great potential from the commercial point of view. Here in this report we demonstrate the urea-to-nitrate molar ratio for various molarities, the heat-treated LiFePO<sub>4</sub>/C powder thus processed displays a well-developed olivine crystalline structure and shows the best electrochemical performance.

## 2. Experimental

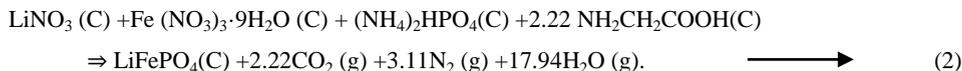
### 2.1 Combustion reactions:

The corresponding chemical reaction is as follows:

- [1]. One mole of NH<sub>2</sub>CONH<sub>2</sub> and NH<sub>2</sub>CH<sub>2</sub>COOH are as fuels, gave four and five moles of the following gases as per the reactions respectively.



- [2]. Theoretical equations for producing LiFePO<sub>4</sub> by using urea are.



In all experiments, the fuel-oxidizer ratio was set to unity; i.e., enough oxidizer was present to fully oxidize the fuel with no required contribution of atmospheric oxygen. To observe the variation in adiabatic flame temperature with fuel-to-nitrate molar ratio, adiabatic flame temperatures were also calculated for nitrate-to-urea molar ratios. The combustion reactions involving these ratios can be presented above equation (1) & (2). Calculation based on thermodynamic considerations help in predicting the exact situations for initiation and ignition of reactions. Adiabatic flame temperature T<sub>ad</sub>, is used to define the temperature at which the enthalpies of the products are equal to those of reactants.

T<sub>ad</sub> can be calculated using the following equations:

The Enthalpy of combustion can be expressed as:

$$\Delta H_{\text{reaction}}^0 = (\sum n \Delta H_f^0)_{\text{products}} - \sum (n \Delta H_f^0)_{\text{reactants}} \longrightarrow \quad (3)$$

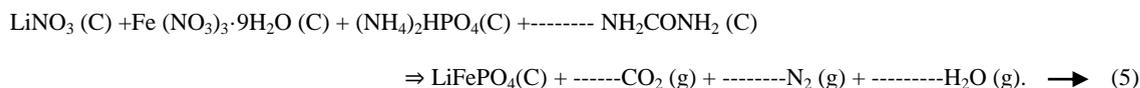
Adiabatic flame temperature can be calculated by using the equation:

$$Q = -\Delta H_{\text{reaction}}^0 = \int_{298}^{T_{\text{ad}}} (\sum n C_p)_{\text{products}} dT \longrightarrow \quad (4)$$

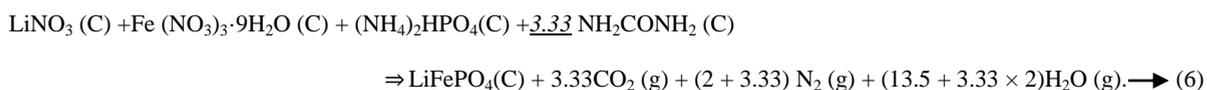
Where n is the number of the mol and T<sub>ad</sub> is the adiabatic flame temperature. Using the thermodynamic data for various reactants and products the enthalpy of combustion and the theoretical adiabatic flame temperatures as a function of urea-to-nitrate molar ratio can be calculated.

### 2.2 Calculation of Fuel- to-Nitrate ratio for Urea:

General chemical formula is:



The corresponding stoichiometric chemical reaction is as follows:



From this chemical reaction nitrate to urea molar ratio is

Nitrates have **3.33 Urea** i.e.,

$$\frac{F}{N} = \frac{\text{Fuel}}{\text{Oxidant}} = \frac{\text{Fuel}}{\text{Nitrate}} = \frac{\text{Urea}}{\text{Nitrate}} = \frac{3.33}{4} = 0.8325 \rightarrow (7)$$

Table 1. Urea-to-Nitrate molar ratio for various molarities

Metal	Fuel	molar ratio	Nitrate/Urea	Enthalpy of combustion	
				$\Delta H_{\text{reaction}}^0$	$T_{\text{ad}}$
Nitrate	Urea	(1:1:1)	0.25	1034.57kcal/mol	0.0302°K
Nitrate	Urea	1:1:2	0.50	490.49kcal/mol	1.277°K
Nitrate	Urea	1:1:3	0.75	-53.57kcal/mol	601°K
Nitrate	Urea	1:1:3.3	0.8250	-216.79kcal/mol	1043°K
Nitrate	Urea	1:1:4	1	-595.85kcal/mol	2001°K
Nitrate	Urea	1:1:5	1.25	-1141.71kcal/mol	2952°K

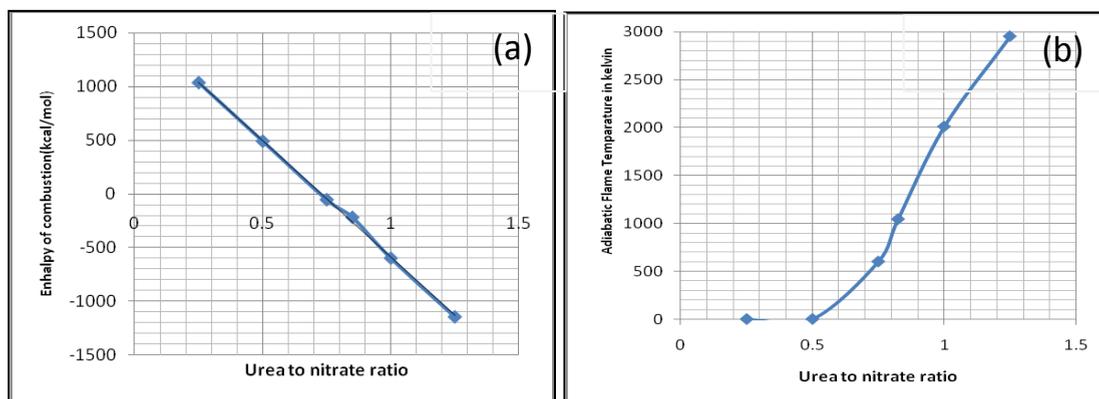


Fig. 1. (a) Enthalpy of combustion (kcal/mol) Vs Urea to nitrate ratio. (b) Adiabatic flame temperature in kelvin Vs Urea to nitrate ratio.

### Synthesis

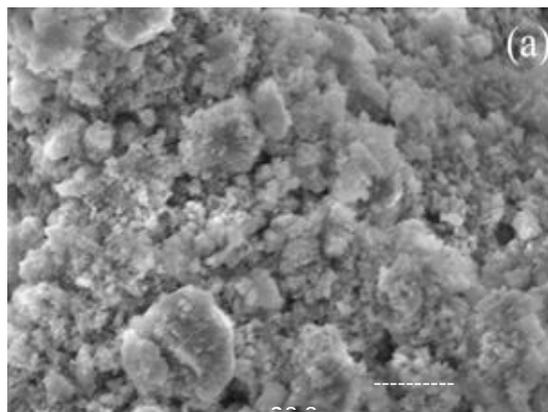
In a typical combustion synthesis the stoichiometric amounts of  $\text{LiNO}_3$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ ,  $(\text{NH}_4)_2\text{HPO}_4$  and urea (ratio of fuel and nitrate) were calculated using the total oxidizing and reducing valencies of the components which provide as numerical coefficients for stoichiometric balance so that the equivalent ratio is unity and the energy



corresponding to  $\text{LiFe}(\text{P}_2\text{O}_7)$ ,  $\text{Li}_3\text{Fe}_2(\text{PO}_4)_3$  and  $\text{Li}_3\text{PO}_4$  in addition to  $\text{LiFePO}_4$ . These impurity phases are formed due to the insufficient quantity of reducing material i.e carbon which is used as carbon source and also may be due to vacuum heat treatment. All the powders showing very fine crystallite sizes. ones reported in literature.

### 3.2. Microstructure

The morphologies of the powders were investigated by a scanning electron microscopy (SEM) (Hitachi S-3400E/N)



**Fig. 3.** SEM images of  $\text{LiFePO}_4/\text{C}$  composites containing samples

S1-Sample urea as fuel with 10% glucose and calcinated at 600C for 8 hrs in 90% Ar and 10% H<sub>2</sub>,

In Fig. 3 shows the scanning electron micrographs of carbon coated  $\text{LiFePO}_4$  which are calcinated in vacuum for 7hrs. The SEM pictures indicate the agglomeration of powder with highly porous nature in all the four cases. From XRD all the powders showing the nano sized crystallite sizes in the range of 40-45nm.

### 3.4. TG–DSC analysis of combustion synthesized powders in static atmosphere:

The synthesized olivine powders were subjected to thermogravimetric/ differential scanning calorimetry (TG-DSC) on a Netzch (Luxx, STA, 449 Jupiter, Germany). Thermogravimetric–differential scanning calorimetry (TG-DSC) was used to establish the calcination temperature for pure  $\text{LiFePO}_4$  phase formation. Fig.4(a) and 4(b) shows the TG–DSC analysis curves for the combustion synthesized carbon coated  $\text{LiFePO}_4$  by using as fuel in static atmosphere at a heating rate of  $10^\circ\text{C}/\text{min}$ . The TG curves indicate one slight mass-loss peak between  $50$  and  $80^\circ\text{C}$ . This mass-loss process is related to endothermic peaks in the DSC curves, due to evaporation of moisture, which is absorbed by the sample. Because these powders are highly hygroscopic. An endothermic peak has been observed at  $436^\circ\text{C}$  for pure  $\text{LiFePO}_4$ , but no appreciable weight loss is observed in the TG curve for the sample. The broad exothermic peak at  $436$ - $601^\circ\text{C}$ , suggesting that the crystallization of  $\text{LiFePO}_4$  takes place at this temperature. The exothermic peaks observed around  $600$ - $700^\circ\text{C}$  correspond to the decomposition of  $\text{LiFePO}_4$  which is formed. Further increasing the temperature these powders melted at the temperature  $889$  and  $890^\circ\text{C}$ . In the case of combustion synthesized powders above  $600^\circ\text{C}$  there is a mass change, due to decomposition of crystalline  $\text{LiFePO}_4$  which is formed below  $600^\circ\text{C}$ .

### TG–DSC analysis of combustion synthesized powders in Nitrogen atmosphere:

Fig.5(a) & 5(b) shows the TG-DSC of the as synthesized powders before calcination. This is performed in nitrogen atmosphere at heating rate of  $10^\circ\text{C}/\text{min}$  by flowing the nitrogen at a rate of  $60\text{ml}/\text{min}$ . Fig.5(a) shows that there is slight mass change in combustion synthesized powders around  $7$  and  $10$  urea respectively. An endothermic peak has been observed at  $436^\circ\text{C}$  for pure  $\text{LiFePO}_4$ , but no appreciable weight loss is observed in the TG curve for sample. The broad exothermic peak at  $436$ - $550^\circ\text{C}$ , suggesting that the crystallization of  $\text{LiFePO}_4$  takes place at this temperature. The exothermic peaks observed around  $600$ - $700^\circ\text{C}$  correspond to the decomposition of  $\text{LiFePO}_4$  which is formed. The endothermic peak around  $756^\circ\text{C}$  in sample corresponding to formation of  $\text{Fe}_2\text{P}$ , which is impurity generally formed in  $\text{LiFePO}_4$ .

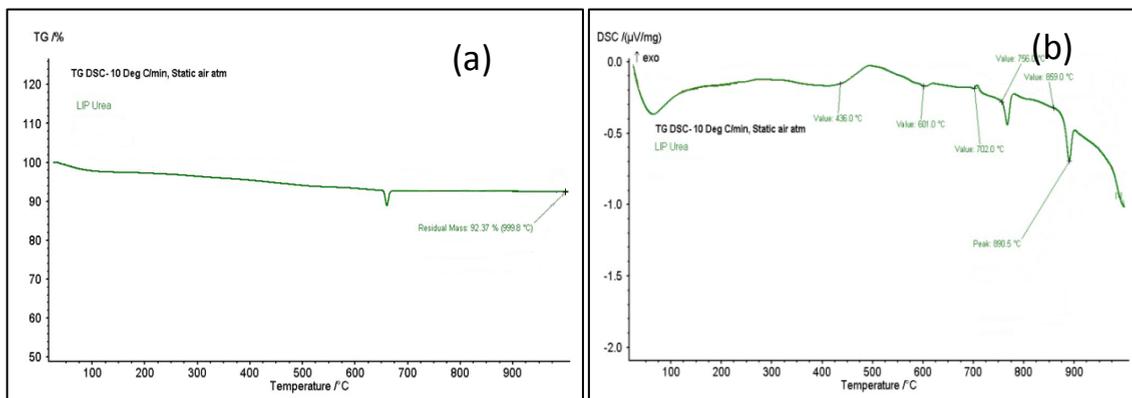


Fig.4.(a). Thermogravimetry of combustion synthesized powders (b). Differential scanning calorimetry of combustion synthesized powders

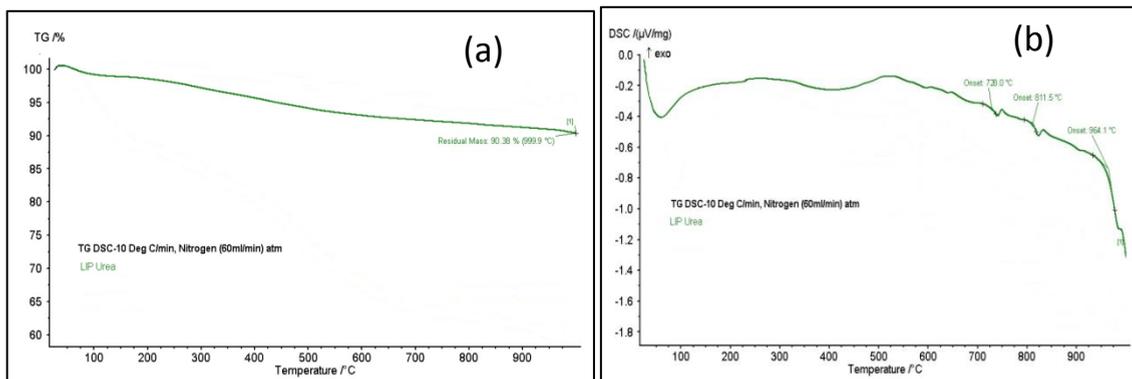


Fig.5 (a). Thermogravimetry of combustion Synthesized powders (b). Differential scanning calorimetry of combustion synthesized powders

3.5. Raman spectroscopy

Raman spectra were obtained from a Horiba Jobin-Yvon Labram HR 800 Raman spectrometer with 514.5 nm Ar-Ion laser radiation at a laser power of 150 mW in the range of 100- 2000cm<sup>-1</sup> for a total of 10 scans to increase

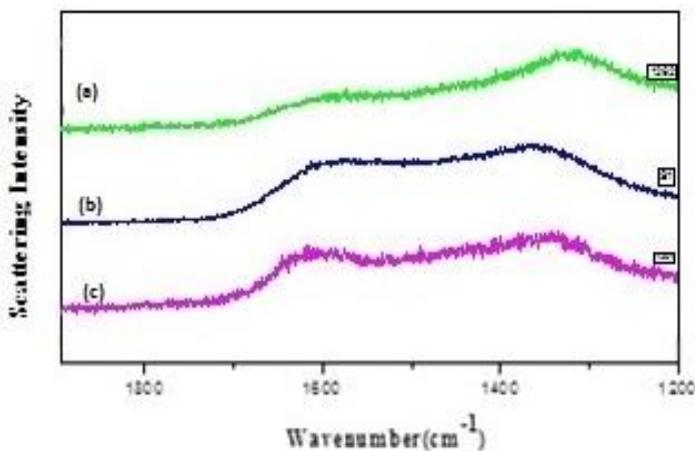


Fig.6 Raman spectrum of six samples (a) Urea+10%Glucose+HT600°C 6h (b)Urea+10%Glucose+VT600°C7h (c) Urea+20%Glucose+VT600°C 7h

the signal-to-noise ratio and with  $4\text{ cm}^{-1}$  spectral resolution. The structure of the surface coated carbon and the material were investigated by Raman spectroscopy. It is well known that the Raman spectra of amorphous carbons exhibit broad bands in the  $1000\sim 1700\text{ cm}^{-1}$ , and the peaks having the maxima in the  $1300\sim 1400\text{ cm}^{-1}$  and  $1550\sim 1650\text{ cm}^{-1}$  are called the D-band and G-band, respectively. The Raman spectra of all samples display two broad peaks at  $1300\sim 1480$  and  $1600\sim 1625\text{ cm}^{-1}$ , both of which were attributed to the D band (disordered carbon,  $\text{sp}^3$ ) and the G band (graphite,  $\text{sp}^2$ ) of Raman vibration modes for amorphous carbon, respectively. Two broad peaks centered at about  $1345$  and  $1620\text{ cm}^{-1}$  corresponds to the D and G bands which show a breathing mode of rings and vibrations of  $\text{sp}^2$  carbon atoms with dangling bonds, respectively.

#### 4. Conclusion

The nanocrystalline  $\text{LiFePO}_4/\text{C}$  nanocomposite material was successfully synthesized by a urea and sucrose modified combustion process, sucrose used as carbon source to increase electronic conductivity and in the initial stages it reduces the ferric to ferrous state. The heat treated samples improves pure and crystallinity of the as prepared sample. The  $\text{LiFePO}_4/\text{C}$  sample shows nano porous morphology with the nano crystallite sizes in the range of  $40\sim 45\text{ nm}$ . With spherical shape morphology, XRD pattern of  $\text{LiFePO}_4/\text{C}$  Nano composite clearly reveals the crystalline orthorhombic. Raman spectra of amorphous carbons exhibit broad bands and the peaks of D-band and G-band, respectively. TG-DSC analysis, An endothermic peak has been observed at  $436^\circ\text{C}$  for pure  $\text{LiFePO}_4$ , but no appreciable weight loss is observed in the TG curve for the sample. The broad exothermic peak at  $436\sim 601^\circ\text{C}$ , suggesting that the crystallization of  $\text{LiFePO}_4$  takes place at this temperature. The exothermic peaks observed around  $600\sim 700^\circ\text{C}$  correspond to the decomposition of  $\text{LiFePO}_4$  which is formed. This can be used as cathode for Lithium ion batteries. It can increase temperature stability of the battery and avoid spilling, boiling or gassing of the batteries. Thus, we can increase safety of the battery. This method is simple, efficient, economical, environment friendly, and has great potent from commercial point of view.

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